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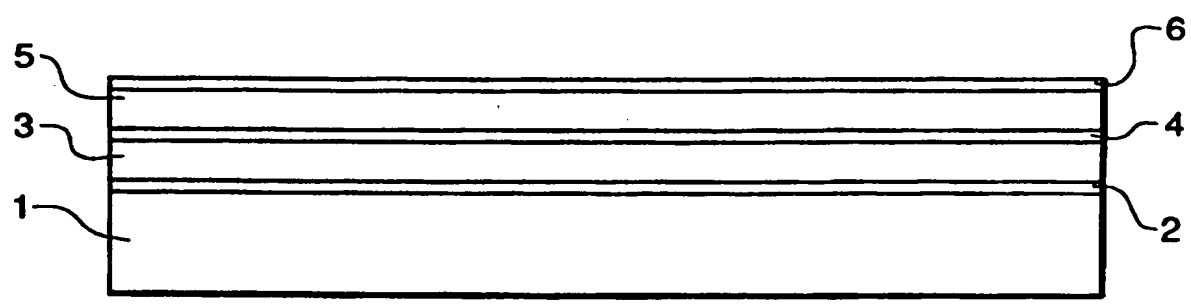
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International Bureau

INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁶ : H01M 10/36, 2/30		A1	(11) International Publication Number: WO 97/19481
			(43) International Publication Date: 29 May 1997 (29.05.97)
(21) International Application Number: PCT/US96/18644			(81) Designated States: AU, BR, CA, FI, JP, KR, MX, NO, RU, SG, UA, European patent (AT, BE, CH, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE).
(22) International Filing Date: 20 November 1996 (20.11.96)			
(30) Priority Data: 08/562,952 24 November 1995 (24.11.95) US			
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Published
With international search report.

(54) Title: A SOLID STATE BATTERY HAVING A DISORDERED HYDROGENATED CARBON NEGATIVE ELECTRODE



(57) Abstract

A solid state battery comprising a substrate (1) at least one multilayered electrochemical cell deposited onto the substrate (1); each multilayered electrochemical cell comprising: a layer of disordered hydrogenated carbon material negative electrode material (3) capable of electrochemically adsorbing and desorbing lithium ions or both lithium and hydrogen ions during charge and discharge; a layer of positive electrode material (5) capable of electrochemically desorbing and adsorbing lithium ions or both lithium and hydrogen ions during charge and discharge; and a layer of insulating/conducting material (4) disposed between the layer of positive electrode material (5) and the layer of negative electrode material (3), where the layer of insulating/conducting material (4) is electrically insulating and capable of readily conducting or transporting lithium ions or both lithium and hydrogen ions from the layer of positive electrode material (5) to the layer of the layer of negative electrode material (3) while the battery is charging and from the layer of negative electrode material (3) to the layer of positive electrode material (5) while the battery is discharging; and an electrically conductive layer (6) deposited atop the last of the at least one multilayered electrochemical cells, the electrically conductive layer (6) providing one battery terminal.

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A SOLID STATE BATTERY HAVING A DISORDERED HYDROGENATED CARBON NEGATIVE ELECTRODE

RELATED APPLICATION INFORMATION

The instant application is a continuation-in-part of U.S. Application Serial No. 08/198,757 filed February 18, 1994, which is in turn a continuation-in-part of U.S. Patent Application Serial No. 08/155,059 filed November 19, 1993, now abandoned.

FIELD OF THE INVENTION

The present invention relates generally to solid state ionic conductors and more specifically to electrically insulating ionic conductors useful as solid state electrolyte and thin-film all solid state batteries employing these ionic conductors.

BACKGROUND OF THE INVENTION

Rechargeable batteries are used in almost every aspect of daily life. A wide variety of industrial, commercial and consumer applications exist. Larger capacity battery uses include such applications as fork lifts, golf carts, uninterruptable power supplies for protection of electronic data storage, and even energy storage for power production facilities. When electric vehicles are manufactured in mass, demand for low weight, high charge capacity batteries will be greater than ever before. Indeed, to make mass use of electric vehicles economically feasible, very high specific capacity may be critically necessary.

In electric vehicles, weight is a significant factor. Because a large component of the total weight of the vehicle is the weight of the batteries, reducing the weight of the cells is a significant consideration in designing batteries to power electric vehicles.

The 1998 California Clean Air Act has posed an exceptional challenge on battery scientists and engineers to develop an improved battery that can support the commercialization of electric vehicles (EV). Needless to say, the law has not changed the reality of battery technology. In over 100 years of rechargeable battery usage, two chemistries namely: Pb-PbO₂ (known as lead-

acid battery) and Cd-NiOOH (known as Ni-Cd battery) have dominate with more than 90% of the market. Neither of the two are likely to fulfill the utopian goals of powering an electric car that will match the range, economy, and performance of an internal combustion engine vehicle. Therefore, battery scientists and engineers are forced to study new battery chemistries.

In addition to industrial, commercial and other large scale uses of batteries, there are literally thousands of consumer applications of rechargeable batteries. A rechargeable electrochemical cell is ideally suited to serve as a portable power source due to its small size, light weight, high power capacity and long operating life. A rechargeable cell may operate as an "install and forget" power source. With the exception of periodic charging, such a rechargeable cell typically performs without attention and rarely becomes the limiting factor in the life of the device it powers.

Present rechargeable battery systems can be classified into two groups those employing liquid electrolytes and those employing solid electrolytes. Liquid electrolyte systems have been around for many decades and are the most well known to the general public. Examples of liquid electrolyte rechargeable battery systems include lead-acid, nickel cadmium, and the more recent nickel-metal hydride systems.

A more recent advancement is the solid electrolyte rechargeable battery systems. The solid electrolyte devices have several distinct advantages over those based on liquid electrolytes. These include (1) the capability of pressure-packaging or hard encapsulation to yield extremely rugged assemblies, (2) the extension of the operating temperature range since the freezing and/or boiling-off of the liquid phase, which drastically affect the device performance when employing liquid electrolytes are no longer a consideration, (3) solid electrolyte devices are truly leak-proof, (4) they have long shelf life due to the prevention of the corrosion of electrodes and of loss of solvent by drying out which occur when using liquid electrolytes, (5) solid electrolytes permit micro-miniaturization, and (6) they do not require heavy, rigid battery cases which are essentially "dead weight" because they provide no additional capacity to the battery but must be included in the total weight thereof.

All of the above considerations have led to a growing use of solid electrolytes. Solid state batteries and timers are already available which employ the solid electrolyte as a cylindrical pellet with suitable electrodes on either side. However, this kind of geometry leads to somewhat poor solid-solid contacts and these devices tend to have high internal resistances and polarization losses. These problems have been overcome by the use of thin films as the electrolytes, since thin films deposited on top of each other have excellent contacts and should also be able to withstand shocks, acceleration forces and spin rates without undue damage.

In forming such a battery system, a solid ion conductor (i.e. solid electrolyte) for moving ions within the system is required. A solid electrolyte is classified by its type of movable ion, such as Li^+ -conductive solid electrolyte, Ag^+ -conductive solid electrolyte, Cu^+ -conductive solid electrolyte, H^+ -conductive solid electrolyte, etc. A solid electrochemical element is constituted by combining one of these solid electrolytes with an appropriate electrode material. Several solid electrolytes are known to exhibit good ionic conductivity, some of which exist in the form of thin films. Oxide ion conductors such as zirconia are operated at high temperatures due to their low conductivity at ambient temperatures. Chloride ion conductors such as PbCl_2 and BaCl_2 have similar temperature restrictions. Silver ion such as AgBr , AgCl , and AgI also show low room temperature ionic conductivity.

Of the thin-film, solid state battery systems, lithium-polymer batteries have received the most widespread interest. Reports in 1979 that lithiated polyethylene-oxide (PEO) possesses lithium ion conductivity raised the expectations for a solid state battery employing PEO as solid electrolyte. Indeed, if PEO, or other polymers, were a true solid electrolyte with practical ionic conductivities and a cationic transfer number of 1, a stable interface with the lithium electrode and good charging uniformity could be realized. The expectations, no doubt, were stimulated by the relative success of the true solid electrolyte "B" Alumina, in the Sodium Sulphur battery.

More recently, several researchers proposed the use of "plasticized polymers" to enhance conductivity at room temperature. Although the term

"plasticized polymers" is the correct material science terminology for the materials, they are in effect no different than a battery separator filled with organic solvent and electrolyte. In this case, we are back to liquid filled systems with all the old unsolved fundamental problems and several new ones.

5 Solid electrolytes consist of solid atomic structures which selectively conduct a specific ion through a network of sites in a two or three dimensional matrix. If the activation energy for mobility is sufficiently low, the solid electrolyte can serve as both the separator and electrolyte in a battery. This can allow one to fabricate an all solid state cell.

10 An important aspect of such electrolytes is that they selectively conduct only one type of ion. If that ion features reversible electrochemistry with both the positive and negative electrode of the battery, and if the solid electrolyte itself is inert to the electrodes, the cell will enjoy a uniform and reversible electrochemistry with no composition change and no passivation or side
15 reactions.

 While true solid electrolyte lithium conductors would not exhibit the inherent problems of Li-polymer systems described herein below, all polymer electrolytes reported to date are not true solid electrolytes. The conductivity occurs in an amorphous zone that conducts anions better than it conducts
20 lithium ions (the transfer number of lithium is less than 0.5). As such, ion concentrations in the electrode surface are variable and irreversible reactions between the anion and the lithium electrodes do occur. The combination of the two effects brings about partial passivation of the lithium surface with non uniform dendritic plating on charge. Additionally, the conductivity of the polymer
25 electrolyte is too low, typically two to four orders of magnitude lower than that of aqueous electrolyte. Also, the electrode area required for a 20 kwh battery is 42 m² for Ni-Cd batteries and is 1610 m² for Li-Polymer batteries. This data clearly conveys that in order to deliver acceptable power levels for EV applications, lithium polymer batteries will require nearly two orders of
30 magnitude, larger electrode area per ampere hour than a higher power density Ni-Cd battery. Given that electrode processing is the most expensive component in battery production and that the cost of electrode processing is

nearly linear with electrode area, the cost implications of the design are astonishing.

In addition to cost, safety of Li batteries, particularly liquid electrolyte systems, is always a problem. The single most important reason rechargeable lithium batteries have not been successful in the market place is their poor safety record. Most research groups that have worked on rechargeable lithium cells have "personally experienced" explosions, and explosions have occurred in the field. The problem can be diagnosed as follows: 1) lithium plating is dendritic, 2) dendrites eventually short through the separator, 3) shorted cells heat up during charging, 4) shorted cells will go into reversal during full battery discharge, 5) low capacity cells will go into reversal during full battery discharge, 6) in reversal, lithium is likely to plate on the cathode which can cause direct chemical reaction between cathode material and lithium, 7) processes 3 and 6 can generate enough heat to melt lithium (165 Centigrade), and 8) molten lithium is an extremely strong reducing agent which will react with most organic and inorganic materials. An explosion could occur depending on: (a) the amount of lithium in the cell, (b) the surface to volume aspect ratio of the cell, (c) the reactivity of the other cell components to lithium, (d) the vapor pressure of the products, and (e) the vent design.

Battery design should be aimed at minimizing the risk of lithium melt down. Given that it is extremely unlikely that lithium melt down can be completely avoided in mass usage of large rechargeable lithium batteries, it is essential to guarantee non explosion when the melt down does occur. Dry polymer electrolyte offers some improvement with regard to exposition when compared to high vapor pressure liquid electrolyte. However, that improvement is counteracted by the need for a very thin separator. Overall, the likelihood of ensuring explosion free melt downs in large cells and batteries is diminutive.

Cells utilizing polymer electrolytes that contain organic solvents, are as likely to be explosive as cells with standard (polymeric) separator and liquid electrolytes. In this case, depending on cell design, common experience places the explosion threshold in the 0.5 to 5 Ah size range; two orders of magnitude smaller than what is required for an EV battery. It should be noted that a

cycled lithium electrode is more prone to explosion than a fresh uncycled one. While this fact has been known for quite some time, lithium polymer battery developers have shied away from publishing safety test data on cycled cells.

In spite of its safety problems, there is a continued interest in lithium

batteries because of their purportedly high power density. This feature makes rechargeable lithium batteries attractive. Theoretical energy densities of most rechargeable lithium chemistries are two and a half to three times higher than that of Pb-Acid and Ni-Cd batteries. Indeed, liquid electrolyte rechargeable lithium batteries could be made to deliver up to 150 Wh/Kg and 200 Wh/liter.

This is about three times higher than the practical gravimetric energy density delivered by the best Ni-Cd batteries and four times higher than the practical gravimetric energy density delivered by the best Pb-Acid batteries. However, the design of the lithium polymer batteries, driven by the poor conductivity of the polymer electrolyte, is very volume inefficient. Specifically, the separator

occupies 30% of the stack volume, carbon is added to the positive electrode in concentration of up to 30% and the positive electrode utilization is poor. Thus, the practical energy density is likely to be considerably lower than of what can be achieved with liquid electrolyte. Estimated deliverable energy density of

lithium polymer batteries is 15-20% of the theoretical energy density. This translates to (using 485 Wh/Kg as theoretical maximum) approximately 70 to 100 Wh/Kg at best. Most likely, compromises that will have to be made to improve manufacturability, safety and cycle life beyond the current laboratory state-of-the-art technology. This will have the effect to reduce the practical

energy density to even below the values proposed above. The power capability of a battery depends upon the physical and chemical properties of the cell components as well as the cell design. Lithium polymer battery developers are trying to counteract the poor inherent conductivity of the polymer electrolytes by reducing the electrode and separator thickness. Because practical

manufacturing reality is likely to impose increases in the electrolyte thickness from approximately 2 to 4 mil, the power deliverable by the cell is likely to drop by 30 to 50%.

An area that requires closer attention is power degradation over life. The main degradation mechanism of the cell involves irreversible reactions between lithium and electrolyte. This reduces the conductivity of the electrolyte as well as increases the impedance of the Lithium electrode due to the formation of passive films; both effects reduce the deliverable power from the battery. Because the cycle life of the lithium polymer battery is short, significant degradation in power is likely to occur in less than 100 cycles.

Other problems arise from real life usage and requirements placed upon battery systems. Traction batteries are assembled from a string of individual cells connected in series. During both charge and discharge, the same amount of current will pass through all the cells. In practical manufacturing and usage, it is impossible to keep all cells at exactly the same state of charge. This forces a weak cell in a battery to go into reverse during deep discharge and some cells to go into overcharge during full charge. For a battery to operate at deep discharge cycles, it is essential that individual cells tolerate reverse or overcharge without damage or safety implications.

Lithium batteries are very poor in this respect. Over discharge will result in plating lithium on the positive electrode which can result in a spontaneous chemical reaction with severe safety implications. Overcharge is likely to result in electrolyte degradation that can generate some volatile gasses as well as increase cell impedance. These problems are particularly severe for lithium cells because: 1) degradation occurs during cycle life, therefore, even if initial capacities are matched very closely, it is unreasonable to expect that the degradation rate will be identical for all cells, 2) the cells tend to develop soft or hard shorts, thereby making it impossible to maintain the cells at the same state of charge at all times, and 3) cell capacity is dependent on temperature, therefore cells that are physically cooler due to their location will deliver less capacity than others. These conditions make the likelihood of cell reversal, relatively early in the life of the battery, very high. Of course, cell reversal is likely to result in venting and or explosion.

It has been propose to install individual diode protection for all cells which could be an expensive, although practical, solution for a portable low

watt-hour battery. The increased cost and reduced reliability associated with this solution makes this very undesirable for an EV battery. Plus, the inherent lack of overcharge and over discharge capability eliminates any possibility of ever developing a rechargeable lithium-polymer battery of a bipolar design.

5 An additional problem with the commercialization of Li-polymer batteries is their high cost. It is difficult to assess the cost, although clearly, processing cost per watt-hour should be much higher than that of traditional batteries. Raw material costs are clearly higher than Pb-Acid, although, it may be similar to Ni-Cd. The cost of raw material will rise due to high purity requirements. There
10 are convincing reasons to expect that lithium polymer batteries, if ever made commercially, will be considerably more expensive than Ni-Cd batteries considering that: 1) primary Li-MnO₂ cells, which are in mass production, are still more expensive than Ni-Cd cells, 2) the purity requirements for a secondary cell are much higher than that of a primary cell, and 3) the electrode area per
15 watt-hour of a lithium polymer secondary battery will be approximately an order of magnitude larger than that of a primary Li-MnO₂ battery.

 Even more problematic than the cost factor is the low cycle life of the lithium polymer batteries, which is particularly important in EV applications. Small rechargeable lithium batteries employing organic liquid electrolyte have
20 delivered 100 to 400 cycles in laboratory tests. It is anticipated that lithium polymer electrolyte batteries of the same size could be made to deliver a comparable number of cycles. However, all the data published to date on lithium polymer batteries was run on cells with a very large amount of excess lithium, therefore, no conclusion can be drawn at this stage.

25 The cycle life of a large multi cell battery is likely to be considerably lower than that of a small two-cell battery. Additional reduction of the expected cycle life results from consideration of the fact that the battery will be limited by the weakest cell, and as previously mentioned, the likelihood of temperature or electrical imbalance is high. Further, power may degrade faster than capacity,
30 so cycle life could become limited due to an unacceptable drop in power. Therefore, it is probably a fair assumption that if a full size battery was built at

today's state-of-the-art technology, it could possibly make 100 cycles or so, which is about an order of magnitude short of what is required for an EV.

Therefore, since lithium-polymer batteries will be inadequate to meet today's requirements for a universally acceptable, thin-film, solid state rechargeable secondary battery system, other solid state systems need to be developed. The solid state battery systems of the present invention meet the requirements discussed hereinabove and provide gravimetric and volumetric energy densities of unparalleled performance.

SUMMARY OF THE INVENTION

There is disclosed herein a thin film, solid state battery. The battery includes a substrate material layer which provides support for the battery and at least one multilayered electrochemical cell deposited onto the substrate. Each cell of the battery includes a layer of negative electrode material, the negative electrode material being capable of electrochemically adsorbing and desorbing ions during charge and discharge thereof, respectively. The multilayered electrochemical cell additionally contains a layer of positive electrode material, the positive electrode material being capable of electrochemically desorbing and adsorbing ions during charge and discharge thereof respectively. Finally the multilayered electrochemical cell contains a layer of insulating/conducting material disposed between the layer of positive electrode material and the layer of negative electrode material. The insulating/conducting material is electrically insulating and capable of readily conducting or transporting ions from the positive electrode to the negative electrode while the battery is charging and from the negative electrode to the positive electrode while the battery is discharging. That battery additionally includes an electrically conductive layer deposited on top of the last of said at least one multilayered electrochemical cells. The electrically conductive material acting as a top battery terminal.

In preferred embodiments, the positive electrode layer includes lithium nickelate or amorphous vanadium oxide and the negative electrode material includes disordered hydrogenated carbon or lithium metal. In another preferred embodiment, one electrode provides a source of lithium ions, while the other

provides a source of hydrogen ions. The solid state lithium or lithium/proton conducting material includes a lithiated or lithiated/hydrogenated electrical insulator material, which may be a lithiated or lithiated/hydrogenated silicon nitride material. The lithiated silicon nitride material preferably has an atomic ratio of between about 20% and about 50% lithium, between about 20% and about 40% silicon and about 20% to about 50 % nitrogen while the lithiated/hydrogenated silicon nitride material preferably has an atomic ratio of between about 10 to 40 atomic % lithium, about 10 to 40 atomic % hydrogen, about 20 to 40 atomic % silicon, and about 20 to 50 atomic % nitrogen.

Preferably the substrate material is formed from an electrically conductive material and acts as one of the electrical terminal of the battery. However, the substrate material may be electrically insulating with an electrically conductive material deposited on it. The deposited electrically conductive material acts as the one of the battery terminals. When the battery includes more than one multi-layered cell, current collector material layers are deposited between the positive electrode of one cell and the negative electrode of the adjacent cell. Typically, the electrically conductive battery terminals and the current collector material layers are formed from non-reactive metals such as molybdenum or aluminum.

BRIEF DESCRIPTION OF THE FIGURES

Figure 1 is a cross-sectional depiction of a first embodiment of the solid state battery of the instant invention specifically illustrating the individual layers thereof;

Figure 2 is a cross-sectional depiction of a second embodiment of the solid state battery of the instant invention specifically illustrating the individual layers thereof, including plural electrochemical cells and current collectors therebetween.

DETAILED DESCRIPTION OF THE INVENTION

Figure 1 is a cross-sectional depiction of a thin-film solid state battery of the present invention. Specifically, reference numeral 1 is the substrate of the

thin-film battery. The substrate provides support for the battery and may also serve as the bottom electrical terminal of the battery. Substrate 1 may be formed from an electrically conductive metal such as aluminum, nickel, copper or stainless steel, or it may be formed from a light weight, electrically insulating polymer or ceramic material. If the substrate 1 is formed of an electrically insulating material or is reactive with the battery electrode materials, then an electrically conductive bottom battery terminal layer 2 is deposited onto the substrate. The material used to form the battery terminal layer 2 may be an electrically conductive metal such as aluminum, nickel or copper, or may even be an electrically conductive ceramic or oxide material. For maximum weight savings, the substrate 1 plus any battery terminal layer 2 should be only as thick as needed to perform their support and conduction functions. Any additional thickness will only increase the "dead weight" of the battery. Typically the total thickness of the substrate 1 plus the battery terminal layer 2 will not be greater than about 200 microns and preferably not greater than about 50 to 100 microns. The battery terminal layer 2 is preferably between 0.5 and 5 microns thick. Deposited on top of the substrate 1 and battery terminal layer 2 is at least one multi-layered electrochemical cell. Each electrochemical cell includes a thin-film negative electrode layer 3, a thin-film positive electrode layer 5 and a thin-film solid electrolyte proton conductive layer 4.

The thin-film negative electrode layer 3 is typically between about 1 and 15 microns thick and is formed from a material which electrochemically adsorbs and desorbs ions such as ionic hydrogen during charging and discharging thereof, respectively. Typically the layer is formed from electrochemical hydrogen storage materials such as metal hydride materials. These metal hydride material may be any of those already known any used in liquid electrolyte nickel-metal hydride batteries. These materials may be AB_2 or AB_5 type metal hydride materials. They may be amorphous, polycrystalline, microcrystalline, nanocrystalline, single crystal or multi-structural materials. They may include only a single compositional phase or may include multiple compositional phases. An extensive review of the known metal hydride

materials useful in electrochemical cells is given in U.S. Patent No. 5,096,667, the disclosure of which is incorporated herein by reference.

In addition to the known metal hydride materials, new metal hydride systems can be developed to take advantage of the environmental differences between an alkaline liquid electrolyte system and the new thin-film solid electrolyte systems. For example, in a liquid electrolyte system, there is generally a problem with corrosion of the electrode due to the caustic nature of the alkaline electrolyte. Therefore, elements which provide corrosion resistance must be added to the negative electrode material to mitigate corrosion damage. In the solid electrolyte system of the present invention, no such corrosion problems will occur due to the absence of caustic liquids and as such, no corrosion inhibitor materials will need to be added to the negative electrode.

Alternatively, in the case of lithium systems, metallic lithium or lithium intercalated disordered hydrogenated carbon can be used as the negative electrode layer 3.

The positive electrode layer 5 is typically between 5 and 20 microns thick and is formed from a material which electrochemically desorbs and adsorbs ions such as ionic hydrogen during charging and discharging thereof, respectively. Typically the layer is formed from a transition metal hydroxide such as nickel hydroxide material. The nickel hydroxide material can be any of those material known in the prior art for use in rechargeable battery systems. They may also be advanced active materials like the locally ordered, disordered, high capacity, long cycle life positive electrode material disclosed in U.S. Patent Application Serial No.s 7/975,031 filed November 12, 1992 and 8/027,973 filed March 8, 1993, the disclosures of which are incorporated herein by reference. These materials include a solid solution nickel hydroxide electrode material having a multiphase structure and at least one compositional modifier to promote said multiphase structure. The multiphase structure comprises at least one polycrystalline γ -phase including a polycrystalline γ -phase unit cell comprising spacedly disposed plates with at least one ion incorporated around the plates, the plates having a range of stable intersheet distances corresponding to a 2^+ oxidation state and a 3.5^+ , or greater, oxidation

state. The compositional modifier is a metal, a metallic oxide, a metallic oxide alloy, a metal hydride, and/or a metal hydride alloy. Preferably the compositional modifier is chosen from the group consisting of Al, Bi, Co, Cr, Cu, Fe, In, LaH₃, Mn, Ru, Sb, Sn, TiH₂, TiO, Zn and mixtures thereof. More preferably, at least three of these compositional modifiers are used. The at least one chemical modifier incorporated is preferably chosen from the group consisting of Al, Ba, Ca, Co, Cr, Cu, F, Fe, K, Li, Mg, Mn, Na, Sr, and Zn.

Also, for lithium ion systems, the positive electrode layer 5 can be formed from a material such as lithium nickelate (LiNiO₄), lithium cobaltate or (LiCoO₄) lithium manganate (LiMnO₄), vanadium oxide, titanium disulfide, etc..

Between the negative electrode layer 3 and the positive electrode layer 5, is deposited a thin-film solid state electrolyte layer 4. This layer is typically between about 0.5 and 2 microns thick, but may be as thin as 1000 Angstroms if the layer onto which it is deposited has a low degree of surface roughness. The type of ionic conductivity required of the solid electrolyte is dependent on the electrochemical reactions involved in the cell. Since the charging cycle electrode reactions of the instant rechargeable protonic battery are:



the solid state electrolyte layer 4 which separates the positive electrode layer 5 and the negative electrode layer 3 must be a proton conductor. That is, the solid electrolyte material must be capable of readily conducting or transporting protons from the positive electrode layer 5 to the negative electrode layer 3 while the battery is charging and from the negative electrode layer 3 to the positive electrode layer 5 while the battery is discharging. The solid electrolyte layer 4 must also be electrically insulating so that the battery electrodes do not short. That is, the electrolyte also acts as the electrode separator. The present inventors have found that a lithiated, lithiated/hydrogenated or hydrogenated electrical insulator has all of the characteristics required depending upon the ion(s) to be transported. Typically this is a lithiated and/or hydrogenated silicon nitride material, but lithiated and/or hydrogenated silicon oxide or lithiated and/or hydrogenated silicon oxynitride may also be used. Preferably the hydrogenated

silicon nitride material has a composition, in atomic percent, of between about 20% and about 50% hydrogen, between about 20% and about 40% silicon and about 20% to about 50 %. The ratio of silicon to nitrogen is generally between about 2:1 and about 1:2, but may be varied outside this range if specifically
5 advantageous under the circumstances.

Alternatively, for the lithium systems, the charging electrode reactions are:



10 therefore, in the lithium systems, a lithium conductor is needed. Solid lithium conductors useful as the ionic conductor layer 4 are lithiated silicon nitride (Li_xSiN_4), lithium phosphate (LiPO_4), lithium titanium phosphate (LiTiPO_4) and lithium phosphonitride ($\text{LiPO}_{4-x}\text{N}_x$ where $0 < x < 1$).

A top battery terminal layer 6 is deposited on top of the positive electrode
15 layer 5. The battery terminal layer 6 is typically between 1 and 5 microns thick and is formed from an electrically conductive material such as a metal or an electrically conductive ceramic or oxide. Specifically, aluminum, copper or nickel may be used.

Turning now to Figure 2 there is depicted therein a solid state battery of
20 the instant invention containing multiple stacked electrochemical cells. The reference numeral of the layers of this battery correspond to those of the battery depicted in Figure 1. Additionally, because this battery includes more than one electrochemical cell, a layer of current collecting material 7 is deposited between positive electrode layer 5 of one cell and the negative electrode layer 3
25 of the adjacent cell. This layer is formed of an electrically conductive material and is typically between 1000 angstroms and 0.5 microns thick. Preferably this layer is formed from a metal such as aluminum, copper or nickel and is resistant to the conduction of protons.

30

EXAMPLE 1

A one square meter multiple cell thin-film solid state battery of the type depicted in Figure 2 having 10 cells will serve as an example of the efficacy of

the present design. Each cell contains a positive electrode layer 5 which is formed from conventional nickel hydroxide and is about 10 microns thick. Each cell also contains a negative electrode layer 3 of metal hydride material and is about 4 microns thick. Finally each cell contains a solid state electrolyte layer 4 formed from hydrogenated silicon nitride material and is about 2 microns thick. Between the cells are current collector layers 7 which are formed of aluminum and are about 0.5 microns thick. The cells are deposited onto an aluminum substrate 1 which also serves as the bottom battery terminal 2. The substrate 1 is about 100 microns thick. On top of the positive electrode layer 5 of the final cell is deposited a top battery terminal layer 6 which is formed of aluminum and is about 5 microns thick.

This battery would have a Specific Capacity calculated as follows:

- 1) Basis 1 m², 1 e⁻ transfer; 10 positive electrode layers formed from Ni(OH)₂
- 2) Density of Ni(OH)₂ = 3.95 g/cm³
- 3) Total volume of 10 Ni(OH)₂ layers = 10*(1m)*(1m)*(10 x 10⁻⁶m) = 1 X 10⁻⁴ m³ or 100 cm³
- 4) Total weight of 10 Ni(OH)₂ layers = (3.95 g/cm³)*(100 cm³) = 395 g Ni(OH)₂
- 5) Charge capacity of Ni(OH)₂ = 289 mAh/g
- 6) Total capacity of 10 Ni(OH)₂ layers = (289 mAh/g)*(395 g) = 114115 mAh = 114.1 Ah
- 7) Charge capacity of metal hydride material = 400 mAh/g
- 8) Weight of metal hydride needed to equal 114.1 Ah = (114.1 ah)*(1g/.400 Ah) = 285 g
- 9) Volume of substrate = (1m)*(1m)*(100 X 10⁻⁶m) = 1 X 10⁻⁴m³ = 100 cm³
- 10) Weight of substrate = (2.7 g/cm³)*(100 cm³) = 270 g
- 11) Total volume of 10 hydrogenated silicon nitride layers = 10*(1m)*(1m)*(2 X 10⁻⁶) = 2 x 10⁻⁵m³ = 20 cm³
- 12) Total weight of 10 hydrogenated silicon nitride layers = (1.7 g/cm³)*(20 cm³) = 34 g

- 13) Total volume of 9 current collector layers = $9 \cdot (1\text{m}) \cdot (1\text{m}) \cdot (0.5 \times 10^{-6}\text{m}) = 4.5 \times 10^{-6}\text{m}^3 = 4.5 \text{ cm}^3$
- 14) Total weight of 9 current collector layers = $(2.7 \text{ g/cm}^3) \cdot (4.5 \text{ cm}^3) = 12.15 \text{ g}$
- 5 15) Volume of top battery terminal = $(1\text{m}) \cdot (1\text{m}) \cdot (5 \times 10^{-6}\text{m}) = 5 \times 10^{-6}\text{m}^3 = 5 \text{ cm}^3$
- 16) Weight of top battery terminal = $(2.7 \text{ g/cm}^3) \cdot (5 \text{ cm}^3) = 13.5 \text{ g}$
- 17) Total battery weight = $(395 \text{ g}) + (285 \text{ g}) + (270 \text{ g}) + (34 \text{ g}) + (12.15 \text{ g}) + (13.5 \text{ g}) = 1009.65 \text{ g} = 1.01 \text{ Kg}$
- 10 18) Specific capacity is $(114.155 \text{ Ah}) / (1.00965 \text{ Kg}) = 113.1 \text{ Ah/Kg}$
- 19) Energy density is $(1.4 \text{ V}) \cdot (113.1 \text{ Ah/Kg}) = \mathbf{158.34 \text{ Wh/Kg}}$
- 20) Volume of battery = $(1\text{m}) \cdot (1\text{m}) \cdot (250 \times 10^{-6}\text{m}) = 2.5 \times 10^{-4}\text{m}^3 = 0.25 \text{ l}$
- 15 21) Volumetric energy density is $(114.155 \text{ Ah}) \cdot (1.4 \text{ V}) / (0.25 \text{ l}) = 639.3 \text{ Wh/l}$

EXAMPLE 2

Another example of the solid state battery having the same structure and dimensions as that in Example 1, but using advanced nickel hydroxide active materials and assuming about 1.7 electron transfer give a specific capacity as calculated below.

- 20 1) Basis 1 m^2 , 1.7 e^- transfer; 10 positive electrode layers formed from advanced $\text{Ni}(\text{OH})_2$ material
- 2) Density of $\text{Ni}(\text{OH})_2 = 3.95 \text{ g/cm}^3$
- 25 3) Total volume of 10 $\text{Ni}(\text{OH})_2$ layers = $10 \cdot (1\text{m}) \cdot (1\text{m}) \cdot (10 \times 10^{-6}\text{m}) = 1 \times 10^{-4} \text{ m}^3$ or 100 cm^3
- 4) Total weight of 10 $\text{Ni}(\text{OH})_2$ layers = $(3.95 \text{ g/cm}^3) \cdot (100 \text{ cm}^3) = 395 \text{ g Ni}(\text{OH})_2$
- 5) Charge capacity of $\text{Ni}(\text{OH})_2 = 483 \text{ mAh/g}$
- 6) Total capacity of 10 $\text{Ni}(\text{OH})_2$ layers = $(483 \text{ mAh/g}) \cdot (395 \text{ g}) = 190785 \text{ mAh} = 190.8 \text{ Ah}$
- 30 7) Charge capacity of metal hydride material = 400 mAh/g

- 8) Weight of metal hydride needed to equal 190.8 Ah = $(190.8 \text{ ah}) \cdot (1 \text{ g} / 400 \text{ Ah}) = 477 \text{ g}$
- 9) Volume of substrate = $(1 \text{ m}) \cdot (1 \text{ m}) \cdot (100 \times 10^{-6} \text{ m}) = 1 \times 10^{-4} \text{ m}^3 = 100 \text{ cm}^3$
- 5 10) Weight of substrate = $(2.7 \text{ g/cm}^3) \cdot (100 \text{ cm}^3) = 270 \text{ g}$
- 11) Total volume of 10 hydrogenated silicon nitride layers = $10 \cdot (1 \text{ m}) \cdot (1 \text{ m}) \cdot (2 \times 10^{-6}) = 2 \times 10^{-5} \text{ m}^3 = 20 \text{ cm}^3$
- 12) Total weight of 10 hydrogenated silicon nitride layers = $(1.7 \text{ g/cm}^3) \cdot (20 \text{ cm}^3) = 34 \text{ g}$
- 10 13) Total volume of 9 current collector layers = $9 \cdot (1 \text{ m}) \cdot (1 \text{ m}) \cdot (0.5 \times 10^{-6} \text{ m}) = 4.5 \times 10^{-6} \text{ m}^3 = 4.5 \text{ cm}^3$
- 14) Total weight of 9 current collector layers = $(2.7 \text{ g/cm}^3) \cdot (4.5 \text{ cm}^3) = 12.15 \text{ g}$
- 15 15) Volume of top battery terminal = $(1 \text{ m}) \cdot (1 \text{ m}) \cdot (5 \times 10^{-6} \text{ m}) = 5 \times 10^{-6} \text{ m}^3 = 5 \text{ cm}^3$
- 16) Weight of top battery terminal = $(2.7 \text{ g/cm}^3) \cdot (5 \text{ cm}^3) = 13.5 \text{ g}$
- 17) Total battery weight = $(395 \text{ g}) + (477 \text{ g}) + (270 \text{ g}) + (34 \text{ g}) + (12.15 \text{ g}) + (13.5 \text{ g}) = 1201.65 \text{ g} = 1.20165 \text{ Kg}$
- 18) Specific capacity is $(190.785 \text{ Ah}) / (1.20165 \text{ Kg}) = 158.8 \text{ Ah/Kg}$
- 20 19) Gravimetric energy density is $(1.4 \text{ V}) \cdot (158.8 \text{ Ah/Kg}) = \mathbf{222.32 \text{ Wh/Kg}}$
- 20) Volume of battery = $(1 \text{ m}) \cdot (1 \text{ m}) \cdot (250 \times 10^{-6} \text{ m}) = 2.5 \times 10^{-4} \text{ m}^3 = 0.25 \text{ l}$
- 21) Volumetric energy density is $(190.785 \text{ Ah}) \cdot (1.4 \text{ V}) / (0.25 \text{ l}) = 1068.4 \text{ Wh/l}$
- 25

EXAMPLE 3

Next, a battery similar to that disclosed in Example 1 except that the protonic system was substituted by a lithium system is presented. Each of the 10 cells contains a positive electrode layer 5 which is formed from lithium nickelate (LiNiO_2) and is about 10 microns thick. Each cell also contains a negative electrode layer 3 of disordered hydrogenated carbon material and is about 4 microns thick. Finally each cell contains a solid state electrolyte layer 4

formed from lithiated silicon nitride material and is about 2 microns thick.

Between the cells are current collector layers 7 which are formed of aluminum and are about 0.5 microns thick. The cells are deposited onto an aluminum substrate 1 which also serves as the bottom battery terminal 2. The substrate 1 is about 100 microns thick. On top of the positive electrode layer 5 of the final cell is deposited a top battery terminal layer 6 which is formed of aluminum and is about 5 microns thick.

This battery would have a Specific Capacity calculated as follows:

- 1) Basis 1 m², 1 e⁻ transfer; 10 positive electrode layers formed from LiNiO₂
- 2) Density of LiNiO₂ = 4.78 g/cm³
- 3) Total volume of 10 LiNiO₂ layers = 10*(1m)*(1m)*(10 x 10⁻⁶m) = 1 X 10⁻⁴ m³ or 100 cm³
- 4) Total weight of 10 LiNiO₂ layers = (4.78 g/cm³)*(100 cm³) = 478 g LiNiO₂
- 5) Charge capacity of LiNiO₂ = 275 mAh/g
- 6) Total capacity of 10 LiNiO₂ layers = (275 mAh/g)*(478 g) = 131450 mAh = 131.5 Ah
- 7) Charge capacity of disordered hydrogenated carbon intercalation material = 370 mAh/g
- 8) Weight of carbon needed to equal 131.5 Ah = (131.5 Ah)*(1g/370 Ah) = 355.3 g
- 9) Volume of substrate = (1m)*(1m)*(100 X 10⁻⁶m) = 1 X 10⁻⁴m³ = 100 cm³
- 10) Weight of substrate = (2.7 g/cm³)*(100 cm³) = 270 g
- 11) Total volume of 10 lithiated silicon nitride layers = 10*(1m)*(1m)*(2 X 10⁻⁶) = 2 x 10⁻⁵m³ = 20 cm³
- 12) Total weight of 10 lithiated silicon nitride layers = (1.7 g/cm³)*(20 cm³) = 34 g
- 13) Total volume of 9 current collector layers = 9*(1m)*(1m)*(0.5 X 10⁻⁶m) = 4.5 X 10⁻⁶m³ = 4.5 cm³

- 14) Total weight of 9 current collector layers = $(2.7 \text{ g/cm}^3) \cdot (4.5 \text{ cm}^3) = 12.15 \text{ g}$
- 15) Volume of top battery terminal = $(1\text{m}) \cdot (1\text{m}) \cdot (5 \times 10^{-6}\text{m}) = 5 \times 10^{-6}\text{m}^3 = 5 \text{ cm}^3$
- 5 16) Weight of top battery terminal = $(2.7 \text{ g/cm}^3) \cdot (5 \text{ cm}^3) = 13.5 \text{ g}$
- 17) Total battery weight = $(478 \text{ g}) + (355.3 \text{ g}) + (270 \text{ g}) + (34 \text{ g}) + (12.15 \text{ g}) + (13.5 \text{ g}) = 1162.95 \text{ g} = 1.16 \text{ Kg}$
- 18) Specific capacity is $(131.450 \text{ Ah}) / (1.16 \text{ Kg}) = 113.0 \text{ Ah/Kg}$
- 19) Energy density is $(3.8 \text{ V}) \cdot (113.0 \text{ Ah/Kg}) = 429.5 \text{ Wh/Kg}$
- 10 20) Volume of battery = $(1\text{m}) \cdot (1\text{m}) \cdot (250 \times 10^{-6}\text{m}) = 2.5 \times 10^{-4}\text{m}^3 = 0.25 \text{ l}$
- 21) Volumetric energy density is $(131.450 \text{ Ah}) \cdot (3.8 \text{ V}) / (0.25 \text{ l}) = 1998.04 \text{ Wh/l}$

15

EXAMPLE 4

Another lithium system battery uses amorphous vanadium oxide as the positive electrode material and lithium metal as the negative electrode. That is, each of the 10 cells contains a positive electrode layer 5 which is formed from amorphous vanadium oxide ($\alpha\text{-V}_2\text{O}_5$) and is about 10 microns thick. Each cell

20 also contains a negative electrode layer 3 of lithium metal which is about 19 microns thick. Finally each cell contains a solid state electrolyte layer 4 formed from lithiated silicon nitride material and is about 2 microns thick. The composition of the lithiated silicon nitride film is typically about 20 to 50 atomic % lithium, about 20 to 40 atomic % silicon, and about 20 to 50 atomic %

25 nitrogen. Between the cells are current collector layers 7 which are formed of molybdenum and are about 0.5 microns thick. The cells are deposited onto an aluminum or nickel substrate 1 which also serves as the bottom battery terminal 2. The substrate 1 is about 50 to 100 microns thick. On top of the positive electrode layer 5 of the final cell is deposited a top terminal layer 6 which is

30 formed of aluminum or molybdenum and is about 5 microns thick.

This battery would have a Specific Capacity calculated as follows:

- 1) Basis 1 m^2 , 1 e^- transfer; 10 positive electrode layers formed from

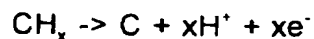
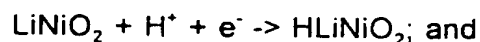
a-V₂O₅

- 2) Density of a-V₂O₅ = 3.3 g/cm³
- 3) Total volume of 10 a-V₂O₅ layers = 10*(1m)*(1m)*(10 x 10⁻⁶m) = 1 X 10⁻⁴ m³ or 100 cm³
- 5 4) Total weight of 10 a-V₂O₅ layers = (3.3 g/cm³)*(100 cm³) = 330 g
- 5) Charge capacity of a-V₂O₅ = 364 mAh/g
- 6) Total capacity of 10 a-V₂O₅ layers = (364 mAh/g)*(330 g) = 120000 mAh = 120 Ah
- 10 7) Volume of lithium 10*(1m)*(1m)*(19 x 10⁻⁶m) = 1.9 X 10⁻⁴ m³ or 190 cm³
- 8) Total weight of 10 lithium layers = 0.53g/cm³ * 190 cm³ = 100.7 g
- 9) Volume of substrate = (1m)*(1m)*(100 X 10⁻⁶m) = 1 X 10⁻⁴m³ = 100 cm³
- 10) Weight of substrate = (2.7 g/cm³)*(100 cm³) = 270 g
- 15 11) Total volume of 10 lithiated silicon nitride layers = 10*(1m)*(1m)*(2 X 10⁻⁶) = 2 x 10⁻⁵m³ = 20 cm³
- 12) Total weight of 10 lithiated silicon nitride layers = (1.7 g/cm³)*(20 cm³) = 34 g
- 20 13) Total volume of 9 current collector layers = 9*(1m)*(1m)*(0.5 X 10⁻⁶m) = 4.5 X 10⁻⁶m³ = 4.5 cm³
- 14) Total weight of 9 current collector layers = (10.2 g/cm³)*(4.5 cm³) = 45.9g
- 15) Volume of top battery terminal = (1m)*(1m)*(5 X 10⁻⁶m) = 5 X 10⁻⁶m³ = 5 cm³
- 25 16) Weight of top battery terminal = (2.7 g/cm³)*(5 cm³) = 13.5 g
- 17) Total battery weight = (330 g)+(100.7 g)+(270 g)+(34 g)+(45.9 g)+(13.5 g) = 794.1 g = 0.794Kg
- 18) Specific capacity is (120 Ah)/(0.794 Kg) = 151.1 Ah/Kg
- 19) Energy density is (3.5 V)*(151.1 Ah/Kg) = **528.9 Wh/Kg**
- 30 20) Volume of battery = (1m)*(1m)*(400 X 10⁻⁶m) = 4 X 10⁻⁴m³ = 0.4 l
- 21) Volumetric energy density is (120 Ah)*(3.5 V)/(0.4 l) = 1050 Wh/l

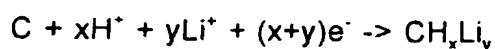
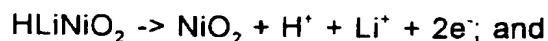
EXAMPLE 5

Finally, a battery combining both protons and lithium ion transfer is presented. Each of the 10 cells contain a positive electrode layer 5 which is formed from lithium nickelate (LiNiO_2) (which can be partially or totally substituted by LiCoO_2 or LiMnO_2) and is about 10 microns thick. Each cell also contains a negative electrode layer 3 of hydrogenated carbon material and is about 4 microns thick. Finally each cell contains a solid state electrolyte layer 4 formed from lithiated/hydrogenated silicon nitride material and is about 2 microns thick. The composition of the lithiated/hydrogenated silicon nitride is typically about 10 to 40 atomic % lithium, about 10 to 40 atomic % hydrogen, about 20 to 40 atomic % silicon, and about 20 to 50 atomic % nitrogen. Between the cells are current collector layers 7 which are formed of aluminum or nickel and are about 0.5 microns thick. The cells are deposited onto an aluminum or nickel substrate 1 which also serves as the bottom battery terminal 2. The substrate 1 is about 100 microns thick. On top of the positive electrode layer 5 of the final cell is deposited a top terminal layer 6 which is formed of aluminum or nickel and is about 5 microns thick. The energy density of this battery should be a combination of the lithium and hydrogen densities. The battery will exhibit multiple charge/discharge plateau voltages corresponding to both lithium and hydrogen cell potentials. This will allow for longer discharge times and increased cell capacity.

This battery can be thought of as a half-charged battery. That is, initially the positive electrode is LiNiO_2 and the negative electrode is hydrogenated carbon. The "as deposited" cell is in a half charged state. During formation (i.e. full charging) hydrogen ions are transferred to the lithium nickelate material according to the following reactions.



The subsequent discharge reactions (i.e. full discharge) are as follows.



Therefore, it can clearly be seen that the solid state batteries of the present invention show tremendous promise for commercial, industrial and consumer uses. Particularly, with regard to the gravimetric and volumetric energy densities shown above, application of these batteries to electric vehicle
5 would be highly advantageous.

It is to be understood that the disclosure set forth herein is presented in the form of detailed embodiments described for the purpose of making a full and complete disclosure of the present invention, and that such details are not to be interpreted as limiting the true scope of this invention as set forth and
10 defined in the appended claims.

We claim:

1. A solid state battery comprising:
a substrate;
at least one multilayered electrochemical cell deposited onto said
5 substrate, each multilayered electrochemical cell comprising:
a layer of disordered hydrogenated carbon negative electrode
material capable of electrochemically adsorbing and desorbing lithium
ions during charge and discharge;
a layer of positive electrode material capable of electrochemically
10 desorbing and adsorbing lithium ions during charge and discharge; and
a layer of insulating/conducting material disposed between said
layer of positive electrode material and said layer of negative electrode
material, where said layer of insulating/conducting material is electrically
insulating and capable of readily conducting or transporting lithium ions
15 from said layer positive electrode material to said layer of negative
electrode material while said battery is charging and from said layer of
negative electrode material to said layer of positive electrode material
while said battery is discharging; and
an electrically conductive layer deposited a top the last of said at least
20 one multilayered electrochemical cells, said electrically conductive layer
providing one battery terminal.
2. The solid state battery of claim 1, wherein said layer of negative
electrode material, said layer of positive electrode material, and said layer of
25 insulating/conducting material are all thin film materials.
3. The solid state battery of claim 1, wherein said layer of positive
electrode material comprises a layer of amorphous V_2O_5 .
- 30 4. The solid state battery of claim 1, wherein said layer of
insulating/conducting material includes a lithiated electrically insulating material.

5. The solid state battery of claim 4, wherein said insulating/conducting material is selected from the group consisting of a lithiated silicon nitride material, a lithiated silicon oxide and a lithiated silicon oxynitride.

5 6. The solid state battery of claim 5, wherein said insulating/conducting material includes a lithiated silicon nitride material.

7. The solid state battery of claim 6, wherein said insulating/conducting material is a lithiated silicon nitride film comprising: about
10 20 to 50 atomic % lithium, about 20 to 40 atomic % silicon, and about 20 to 50 atomic % nitrogen.

8. The solid state battery of claim 1, wherein said substrate is electrically conductive and acts as a second electrical terminal of said battery.
15

9. The solid state battery of claim 1, wherein said substrate is electrically insulative and an electrically conductive material layer is deposited onto said substrate, said electrically conductive material layer acting as a second battery terminal.
20

10. The solid state battery of claim 1, including more than one of said multilayered electrochemical cells and further comprising current collector material layers deposited between said layer of positive electrode material of one multilayered electrochemical cell and said layer of negative electrode
25 material of an adjacent electrochemical cell.

11. A solid state battery comprising:
a substrate;
at least one multilayered electrochemical cell deposited onto said
30 substrate, each layer of said multilayered electrochemical cell comprising:

a layer of disordered hydrogenated carbon negative electrode material capable of electrochemically adsorbing and desorbing both protons and lithium ions during charge and discharge;

5 a layer of positive electrode material capable of electrochemically desorbing and adsorbing both protons and lithium ions during charge and discharge; and

a layer of insulating/conducting material disposed between said layer of positive electrode material and said layer of negative electrode material, where said layer of insulating/conducting material is electrically
10 insulating and capable of readily conducting or transporting both protons and lithium ions from said layer positive electrode material to said layer of negative electrode material while said battery is charging and from said layer of negative electrode material to said layer of positive electrode material while said battery is discharging; and

15 an electrically conductive layer deposited a top the last of said at least one multilayered electrochemical cells, said electrically conductive layer providing one battery terminal.

20 12. The solid state battery of claim 11, wherein said layer of negative electrode material, said layer of positive electrode material, and said layer of insulating/conducting material are all thin film materials.

25 13. The solid state battery of claim 11, wherein said layer of positive electrode material comprises a layer of at least one of LiNiO_2 , LiCoO_2 , or LiMnO_2 .

30 14. The solid state battery of claim 11, wherein said layer of insulating/conducting material includes a lithiated/hydrogenated electrically insulating material.

15. The solid state battery of claim 11, wherein said insulating/conducting material is selected from the group consisting of a

lithiated/hydrogenated silicon nitride material, a lithiated/hydrogenated silicon oxide and a lithiated/hydrogenated silicon oxynitride.

16. The solid state battery of claim 15, wherein said
5 insulating/conducting material includes a lithiated/hydrogenated silicon nitride material.

17. The solid state battery of claim 16, wherein said
insulating/conducting material is a lithiated/hydrogenated silicon nitride film
10 comprising: about 10 to 40 atomic % lithium, about 10 to 40 atomic %
hydrogen, about 20 to 40 atomic % silicon, and about 20 to 50 atomic %
nitrogen.

18. The solid state battery of claim 11, wherein said substrate is
15 electrically conductive and acts as a second electrical terminal of said battery.

19. The solid state battery of claim 11, wherein said substrate is
electrically insulative and an electrically conductive material layer is deposited
onto said substrate, said electrically conductive material layer acting as a
20 second battery terminal.

20. The solid state battery of claim 11, including more than one of said
multilayered electrochemical cells and further comprising current collector
material layers deposited between said layer of positive electrode material of
25 one multilayered electrochemical cell and said layer of negative electrode
material of an adjacent multilayered electrochemical cell.

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FIG - 1

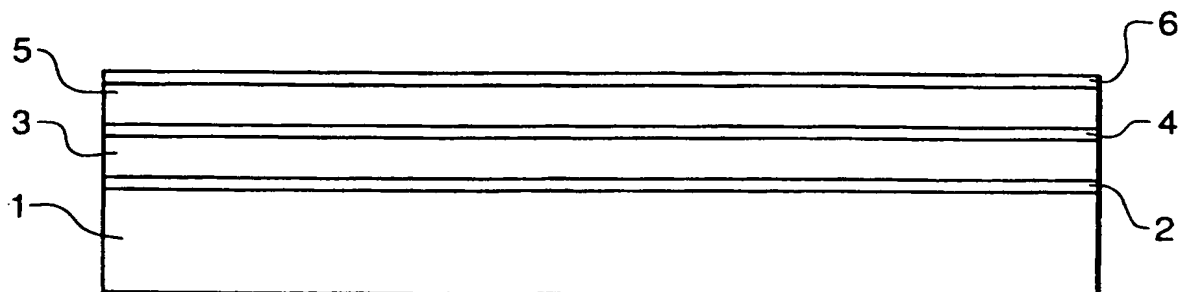
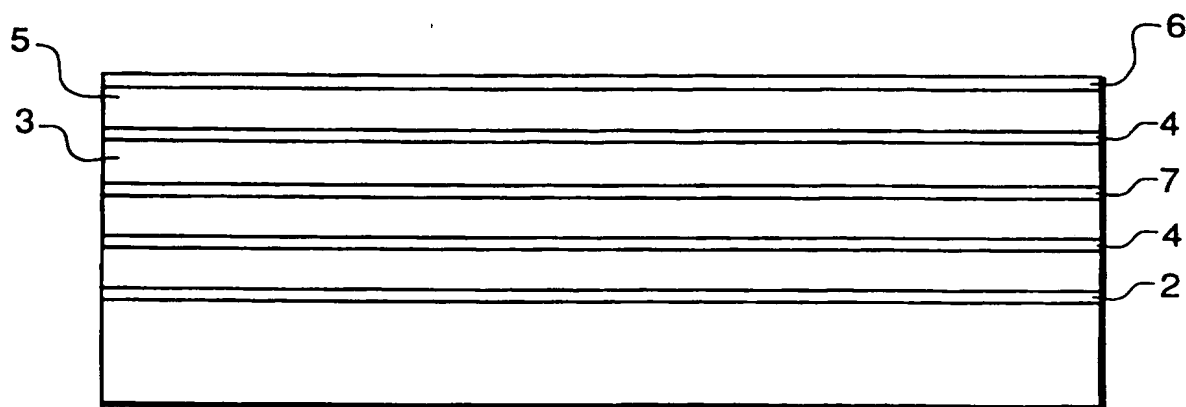


FIG - 2



INTERNATIONAL SEARCH REPORT

International application No.
PCT/US96/18644**A. CLASSIFICATION OF SUBJECT MATTER**

IPC(6) :H01M 10/36, 2/30

US CL :429/152, 162, 178, 191

According to International Patent Classification (IPC) or to both national classification and IPC

B. FIELDS SEARCHED

Minimum documentation searched (classification system followed by classification symbols)

U.S. : 429/152, 162, 178, 191, 59

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched
NONEElectronic data base consulted during the international search (name of data base and, where practicable, search terms used)
NONE**C. DOCUMENTS CONSIDERED TO BE RELEVANT**

Category*	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
X	US 4,810,599 A (KONDO ET AL) 07 March 1989, col. 1, lines 10-27; col. 3, lines 22-col. 4, line 13; Figs. 1,2	1-5, 8-10

☐ Further documents are listed in the continuation of Box C. ☐ See patent family annex.

* Special categories of cited documents:	* T	later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention
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Date of the actual completion of the international search 06 JANUARY 1997	Date of mailing of the international search report 05 FEB 1997
Name and mailing address of the ISA/US Commissioner of Patents and Trademarks Box PCT Washington, D.C. 20231 Facsimile No. (703) 305-3230	Authorized officer JOHN S. MAPLES Telephone No. (703) 308-1795

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